

MAT

**BIAXIAL CREEP-FATIGUE BEHAVIOR  
OF MATERIALS FOR SOLAR THERMAL SYSTEMS**

by

**S. Majumdar**



---

**ARGONNE NATIONAL LABORATORY, ARGONNE, ILLINOIS**

**Prepared for the U. S. DEPARTMENT OF ENERGY**

**under Contract W-31-109-Eng-38**

The facilities of Argonne National Laboratory are owned by the United States Government. Under the terms of a contract (W-31-109-Eng-38) among the U. S. Department of Energy, Argonne Universities Association and The University of Chicago, the University employs the staff and operates the Laboratory in accordance with policies and programs formulated, approved and reviewed by the Association.

#### MEMBERS OF ARGONNE UNIVERSITIES ASSOCIATION

The University of Arizona	The University of Kansas	The Ohio State University
Carnegie-Mellon University	Kansas State University	Ohio University
Case Western Reserve University	Loyola University of Chicago	The Pennsylvania State University
The University of Chicago	Marquette University	Purdue University
University of Cincinnati	The University of Michigan	Saint Louis University
Illinois Institute of Technology	Michigan State University	Southern Illinois University
University of Illinois	University of Minnesota	The University of Texas at Austin
Indiana University	University of Missouri	Washington University
The University of Iowa	Northwestern University	Wayne State University
Iowa State University	University of Notre Dame	The University of Wisconsin-Madison

#### NOTICE

This report was prepared as an account of work sponsored by an agency of the United States Government. Neither the United States Government or any agency thereof, nor any of their employees, make any warranty, express or implied, or assume any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represent that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, mark, manufacturer, or otherwise, does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.

Printed in the United States of America  
Available from  
National Technical Information Service  
U. S. Department of Commerce  
5285 Port Royal Road  
Springfield, VA 22161

NTIS price codes  
Printed copy: A03  
Microfiche copy: A01

Distribution Category:  
Solar Thermal--Large Scale Systems  
(UC-62c)

---

ANL-80-34

---

ARGONNE NATIONAL LABORATORY  
9700 South Cass Avenue  
Argonne, Illinois 60439

BIAXIAL CREEP-FATIGUE BEHAVIOR  
OF MATERIALS FOR SOLAR THERMAL SYSTEMS\*

by

S. Majumdar

Materials Science Division

May 1980

\*Work supported by the U. S. Department of Energy under Sandia Contract  
No. 85339 for the Solar Thermal Program.

TABLE OF CONTENTS

	<u>Page</u>
ABSTRACT . . . . .	1
I. INTRODUCTION . . . . .	1
II. EXPERIMENTAL DETAILS . . . . .	2
III. TEST RESULTS . . . . .	5
IV. DISCUSSION . . . . .	14
V. CONCLUSIONS . . . . .	27
ACKNOWLEDGMENTS . . . . .	28
REFERENCES . . . . .	28

## LIST OF FIGURES

<u>No.</u>	<u>Title</u>	<u>Page</u>
1.	Typical Microstructure of (a) Type 316H Stainless Steel and (b) Incoloy 800 in the As-received Condition . . . . .	4
2.	Hysteresis Loops for Incoloy 800 at an Axial Strain Rate of $1 \times 10^{-3}$ /s at 1100°F . . . . .	8
3.	Hysteresis Loops for Incoloy 800 at an Axial Strain Rate of $4 \times 10^{-3}$ /s at 1100°F . . . . .	8
4.	Hysteresis Loops for Type 316H Stainless Steel at an Axial Strain Rate of $4 \times 10^{-3}$ /s at 1100°F . . . . .	9
5.	Hysteresis Loops for Test 1070 . . . . .	9
6.	Hysteresis Loops for Test 1069 . . . . .	10
7.	Hysteresis Loops for Test 1091 . . . . .	10
8.	Hysteresis Loops for Test 1079 . . . . .	11
9.	Hysteresis Loops for Test 1075 . . . . .	11
10.	Hysteresis Loops for Test 1061 . . . . .	12
11.	Hysteresis Loops for Test 1082 . . . . .	12
12.	Hysteresis Loops for Type 316H Stainless Steel at Room Temperature . . . . .	13
13.	Stress-relaxation Behavior of Incoloy 800 and Type 316H Stainless Steel at 1100°F . . . . .	14
14.	Axial Stress Range Versus Cycle Plots for Continuous-cycling Fatigue with Zero Internal Pressure at 1100°F and at $\Delta\epsilon_t = 0.5\%$ . . . . .	15
15.	Axial Stress Range Versus Cycle Plots for Continuous-cycling Fatigue with Internal Pressures at 1100°F and at $\Delta\epsilon_t = 0.5\%$ . . . . .	15
16.	Axial Stress Range Versus Cycles for 1-min Compressive Hold Tests with Zero Internal Pressure at 1100°F and $\Delta\epsilon_t = 0.5\%$ . . . . .	16
17.	Axial Stress Range Versus Cycles for 1-min Compressive Hold Tests with Internal Pressure at 1100°F and $\Delta\epsilon_t = 0.5\%$ . . . . .	17

## LIST OF FIGURES (CONTD.)

<u>No.</u>	<u>Title</u>	<u>Page</u>
18.	Axial Stress Range Versus Cycles for 1-min Tensile Hold Tests at 1100°F and $\Delta\epsilon_t = 0.5\%$ . . . . .	17
19.	Hysteresis Loops for 1-min Tensile Hold and 1-min Compressive Hold Tests of Type 316H Stainless Steel at 1100°F . . . . .	18
20.	Variation of Mean Diametral Strain with Cycles for Continuous-cycling Fatigue with Internal Pressure at 1100°F and $\Delta\epsilon_t = 0.5\%$ . . . . .	21
21.	Variation of Mean Diametral Strain with Cycles for 1-min Compressive Hold-time Tests with Internal Pressure at 1100°F and $\Delta\epsilon_t = 0.5\%$ . . . . .	21
22.	Variation of Mean Diametral Strain with Cycles for 1-min Tensile-hold Tests with an Internal Pressure of 1100 psi at 1100°F and $\Delta\epsilon_t = 0.5\%$ . . . . .	22
23.	Number of Initiation Sites for Incoloy 800 Specimens under Continuous-cycling Fatigue at 1100°F . . . . .	22
24.	Scanning Electron Micrographs Showing Fracture and ID Surfaces of Incoloy 800 specimens from Two Biaxial Creep-fatigue Tests . . . . .	24
25.	Scanning Electron Micrograph of Fracture Surface from Test No. 1041 . . . . .	25
26.	Scanning Electron Micrographs of the Fracture Surface of an Incoloy 800 Specimen Subjected to 1-min Tensile Hold in Test 1066 . . . . .	26
27.	Scanning Electron Micrographs of the Fracture Surface of an Incoloy 800 Specimen Subjected to 1-min Tensile Hold in Test 1069 . . . . .	26

## LIST OF TABLES

<u>No.</u>	<u>Title</u>	<u>Page</u>
I.	Composition of Incoloy 800 Seamless Tubing . . . . .	3
II.	Nominal Room-temperature Mechanical Properties of Type 316H Stainless Steel and Incoloy 800 Tubing . . . . .	3
III.	Summary of Biaxial Creep-fatigue Tests of Type 316H Stainless Steel . . . . .	6
IV.	Summary of Biaxial Creep-fatigue Tests for Incoloy 800 . . . . .	7
V.	Summary of Relaxation of Stresses at Half-life for the 1-min Hold-time Tests on Incoloy 800 at 593°C and $\Delta\epsilon_t = 0.5\%$ . . . . .	19
VI.	Summary of Relaxation of Stresses at Half-life for the 1-min Hold-time Tests on Type 316H Stainless Steel at 593°C and $\Delta\epsilon_t = 0.5\%$ . . . . .	20

# BIAXIAL CREEP-FATIGUE BEHAVIOR OF MATERIALS FOR SOLAR THERMAL SYSTEMS

by

S. Majumdar

## ABSTRACT

Biaxial creep-fatigue data for Incoloy 800 and Type 316H stainless steel at elevated temperature are presented. Tubular specimens were subjected to constant internal pressure and strain-controlled axial cycling with and without hold times in tension as well as in compression. The results show that the internal pressure affects diametral ratchetting and axial stress range significantly. However, the effect of a relatively small and steady hoop stress on the cyclic life of the materials is minimal. A 1-min compressive hold per cycle does not seriously reduce the fatigue life of either material; a tensile hold of equal duration causes a significant reduction in life for Type 316H stainless steel, but none for Incoloy 800. Fracture surfaces of specimens made of both materials were studied by scanning electron microscopy to determine the reason for the difference in behavior.

## I. INTRODUCTION

A general feature of solar thermal systems that is distinctly different from the operating conditions associated with fossil and nuclear power plants is the highly cyclic nature of the thermal loading experienced by critical components. Solar thermal systems will undergo at least one major start-up and shutdown cycle per day, with additional cycles likely to be imposed by intermittent cloud cover and unscheduled maintenance and repair. Thus, critical components may be expected to accumulate of the order of tens of thousands of cycles over their design lifetime. In many cases, such as the solar central receiver, the temperatures and stresses will be sufficiently high to introduce creep-fatigue-environment interaction as a major life-limiting factor. A further complicating factor in many solar thermal systems is the highly asymmetric nature of the thermal load, which together with the pressure load often results in the creation of a multi-axial state of stress in critical components of a solar system. Unfortunately, virtually no multi-axial creep-fatigue data are currently available for any material.



The present program was initiated in order to address the problem of creep-fatigue under a biaxial state of stress. The materials chosen were Type 316H stainless steel and Incoloy 800, both of which are candidate materials for use in solar thermal systems. Tubular specimens were subjected to a constant internal pressure and strain-controlled axial cycling with and without hold time at elevated temperature. The data generated for Type 316H stainless steel have been published in detail in a previous report.<sup>1</sup> The present report summarizes the results obtained for Incoloy 800 and compares the observed behavior with that of Type 316H stainless steel.

## II. EXPERIMENTAL DETAILS

Details of the specimen design and test equipment were described in Ref. 1. Specimens were fabricated from 1-in.-diameter seamless tubing supplied by Pacific Tube Company of Los Angeles, California; tube dimensions were 1-in. OD x 0.109-in. (min) wall for Type 316H stainless steel and 1-in. OD x 0.125-in. wall for Incoloy 800. Chemical analysis of the Type 316H stainless steel was described in Ref. 1; similar data for Incoloy 800 are given in Table I. The Incoloy 800 tubing was given an annealed and pickled finish by the vendor and satisfied ASME specification SB-163. All the specimens were tested in the as-received condition. Nominal room-temperature mechanical properties of both the materials, as supplied by the vendor, are given in Table II. Micrographs of the as-received materials, shown in Fig. 1, indicate that the grain structures are generally equiaxed with average ASTM grain sizes of 6.5 and 6.3 in transverse section and 6.4 and 5.9 in longitudinal section for Type 316H stainless steel and Incoloy 800, respectively. Note that the grain size for the Incoloy 800 material is rather large and consequently the present heat may not be representative of an average heat of Incoloy 800.

The biaxial fatigue testing was carried out in a closed-loop servo-controlled MTS testing machine using constant internal pressure and axial strain control. The internal pressure was provided by commercially available pressurized air bottles. Axial and diametral strains were measured by means of high-temperature extensometers and the axial load was measured by a 40-kips load cell. The specimen was heated by a Lepel induction heater operating at a frequency of 455 kHz. The maximum temperature variation in the central 0.5-in. gauge length of the specimen was  $\pm 10^{\circ}\text{F}$ .

The test procedure consisted of first heating the specimen to the desired temperature with zero axial load, and holding the temperature steady until the whole system came to thermal equilibrium. The internal pressure, if any, was then applied and the specimen kept at the temperature for sufficient time to allow the new temperature distribution to reach equilibrium. The specimen was then cycled axially under axial strain control. Hysteresis loops of axial stress versus axial strain and axial strain versus diametral strain were recorded on x-y plotters at regular intervals. Each individual signal was also plotted on a strip-chart recorder. For the internally pressurized specimens, the test was shut down automatically when a crack penetrated through the wall. For the unpressurized specimens, the test was shut